

CHAPTER 4: IMPERFECTIONS IN SOLIDS

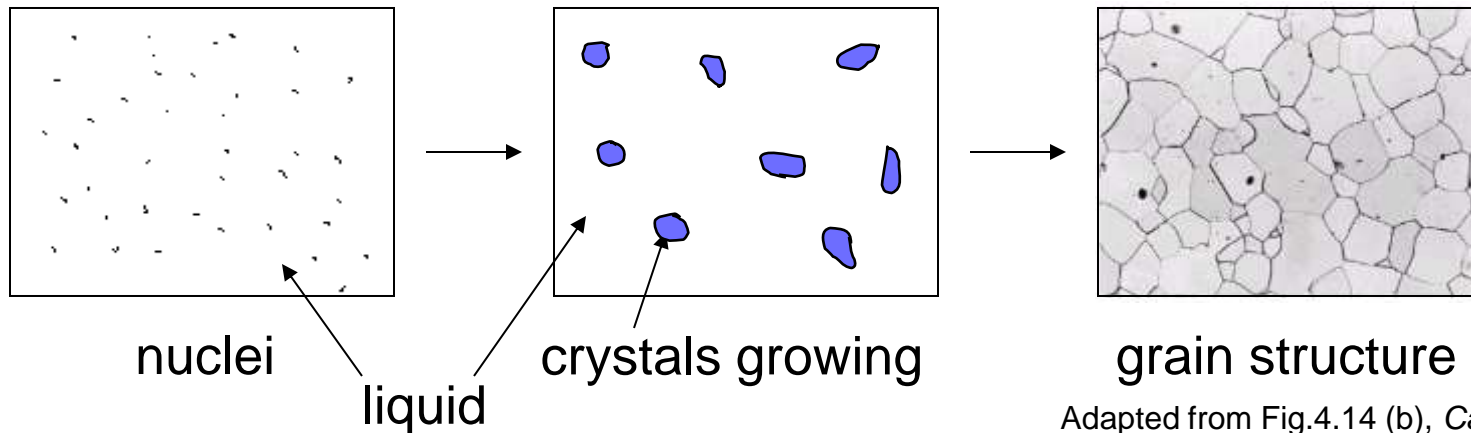
ISSUES TO ADDRESS...

- What are the solidification mechanisms?
- What types of defects arise in solids?
- Can the number and type of defects be varied and controlled?
- How do defects affect material properties?
- Are defects undesirable?



Imperfections in Solids

- **Solidification**- result of casting of molten material
 - 2 steps
 - Nuclei form
 - Nuclei grow to form crystals – grain structure
- Start with a molten material – all liquid



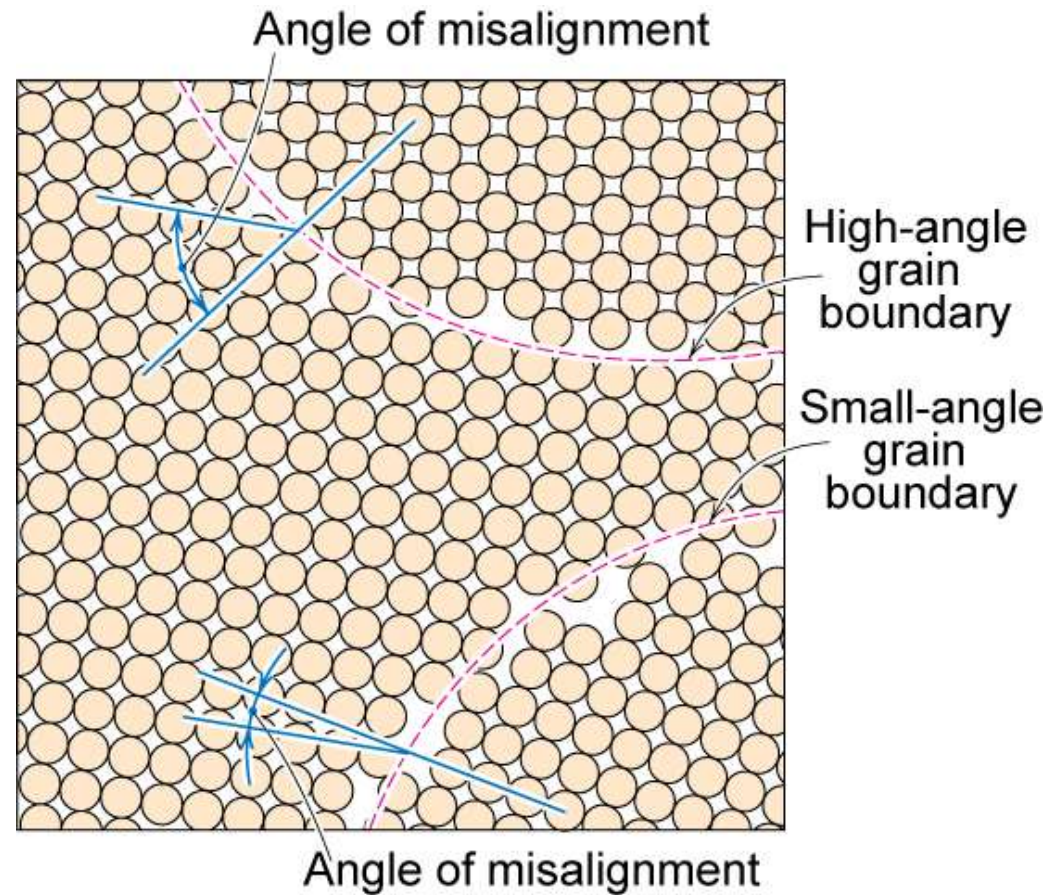
Adapted from Fig.4.14 (b), *Callister 7e*.

- Crystals grow until they meet each other

Polycrystalline Materials

Grain Boundaries

- regions between crystals
- transition from lattice of one region to that of the other
- slightly disordered
- low density in grain boundaries
 - high mobility
 - high diffusivity
 - high chemical reactivity



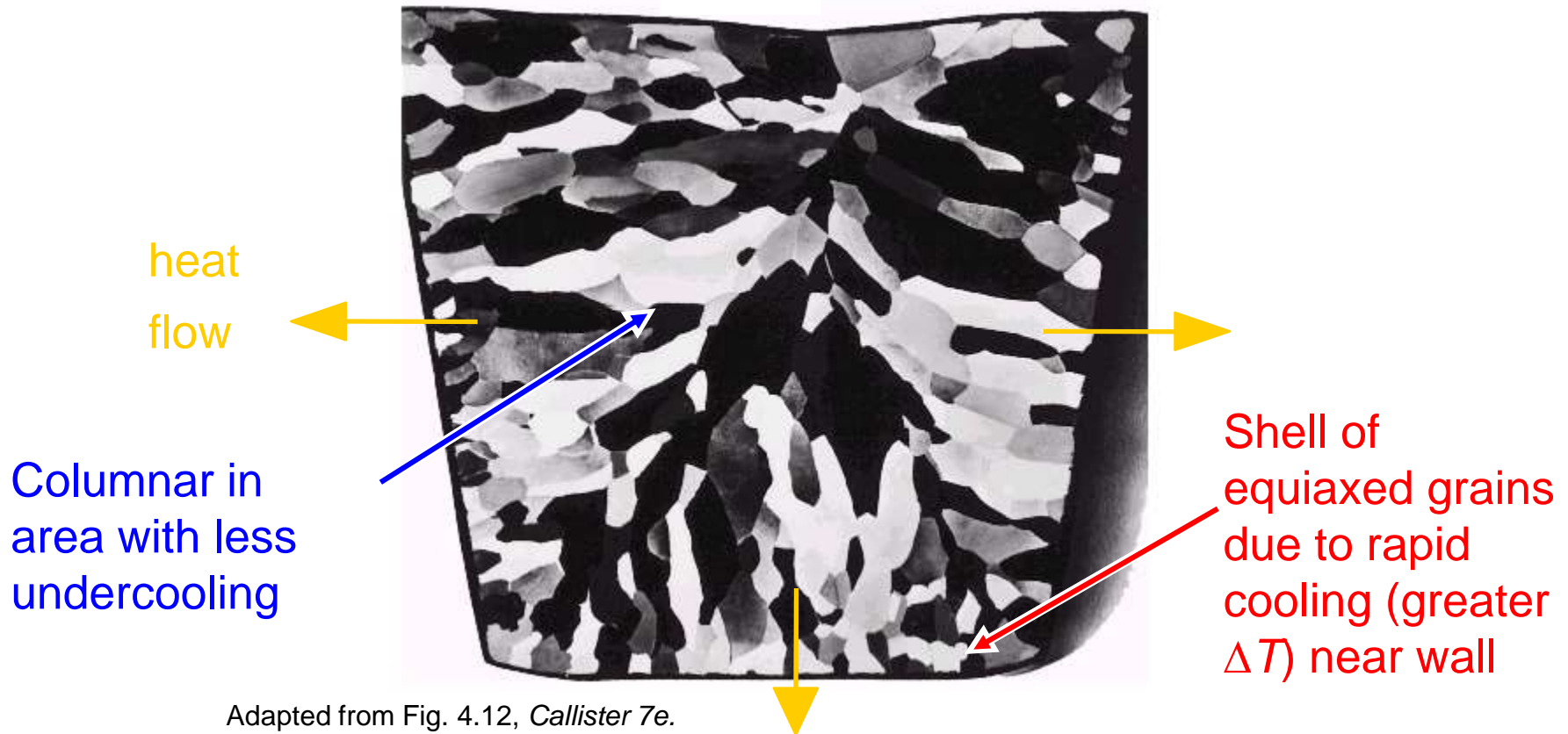
Adapted from Fig. 4.7, *Callister 7e*.



Solidification

- Grains can be
- equiaxed (roughly same size in all directions)
 - columnar (elongated grains)

← ~ 8 cm →



Adapted from Fig. 4.12, *Callister 7e*.

Grain Refiner - added to make smaller, more uniform, equiaxed grains.

Imperfections in Solids

There is no such thing as a perfect crystal.

- What are these imperfections?
- Why are they important?

Many of the important properties of materials are due to the presence of imperfections.



Types of Imperfections

- Vacancy atoms
- Interstitial atoms
- Substitutional atoms

Point defects

- Dislocations

Line defects

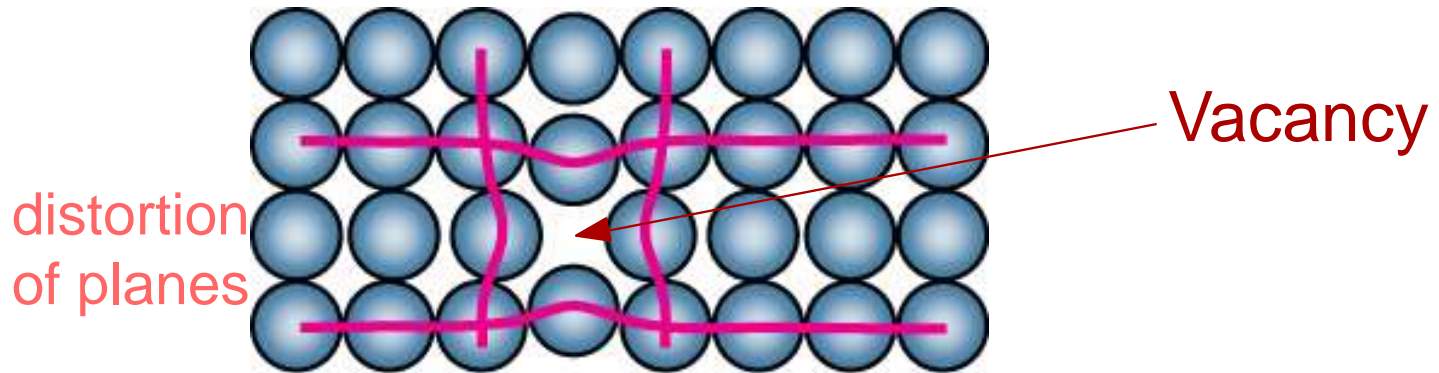
- Grain Boundaries

Area defects

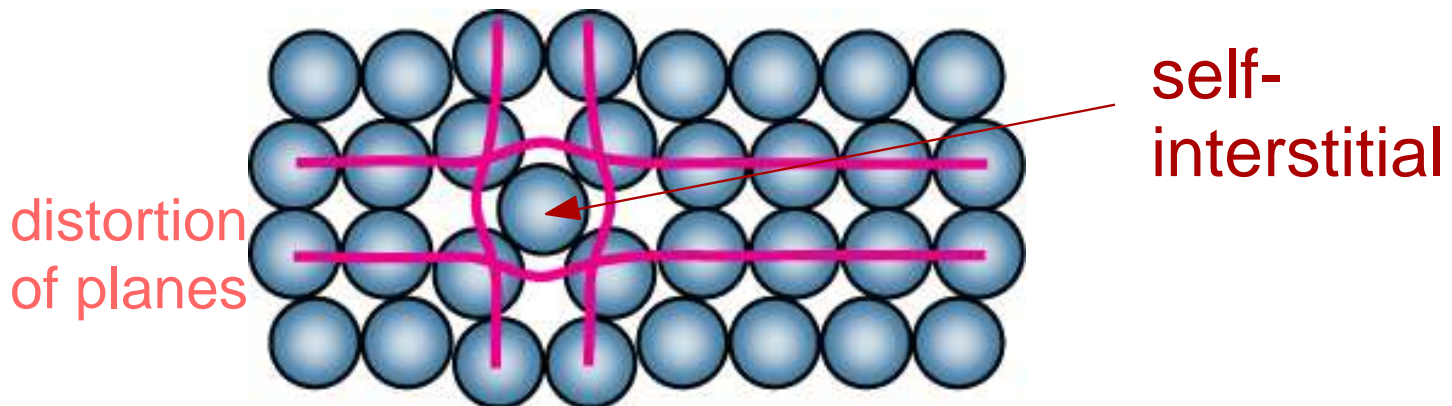


Point Defects

- **Vacancies:**
-vacant atomic sites in a structure.



- **Self-Interstitials:**
-"extra" atoms positioned between atomic sites.



Equilibrium Concentration: Point Defects

- Equilibrium concentration varies with temperature!

No. of defects $\rightarrow N_v$

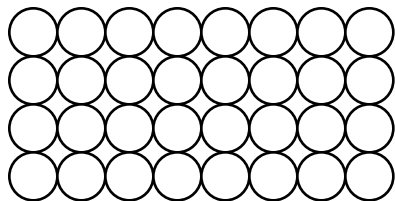
No. of potential defect sites. $\rightarrow N$

Activation energy $\rightarrow -Q_v$

Boltzmann's constant $\rightarrow k$

Temperature $\rightarrow T$

$$\frac{N_v}{N} = \exp\left(\frac{-Q_v}{kT}\right)$$



Each lattice site
is a potential
vacancy site

$$(1.38 \times 10^{-23} \text{ J/atom-K})$$
$$(8.62 \times 10^{-5} \text{ eV/atom-K})$$

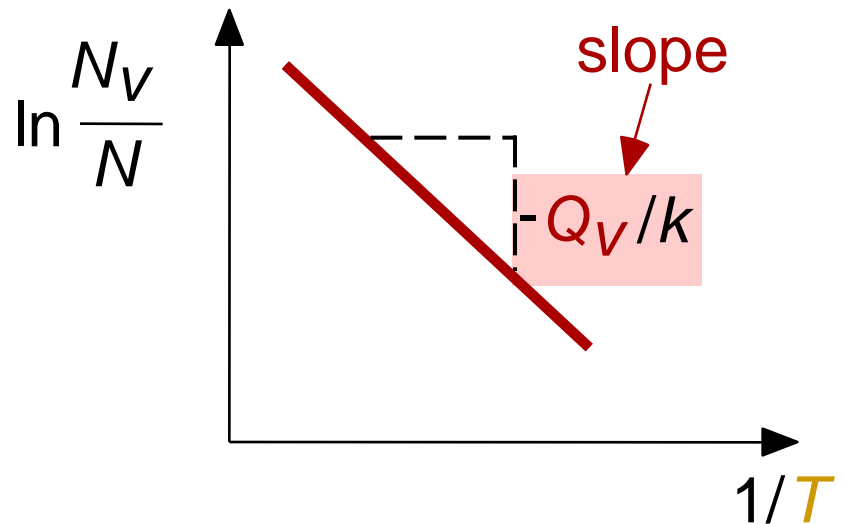
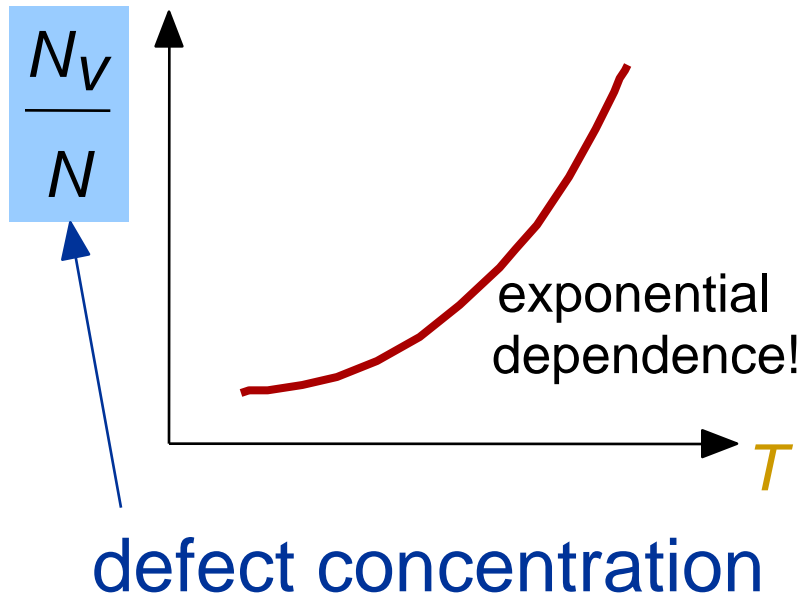


Measuring Activation Energy

- We can get Q_v from an experiment.
- Measure this...

$$\frac{N_v}{N} = \exp\left(\frac{-Q_v}{kT}\right)$$

- Replot it...



Estimating Vacancy Concentration

- Find the equil. # of vacancies in 1 m³ of Cu at 1000°C.
- Given:

$$\rho = 8.4 \text{ g/cm}^3 \quad A_{\text{Cu}} = 63.5 \text{ g/mol}$$

$$Q_V = 0.9 \text{ eV/atom} \quad N_A = 6.02 \times 10^{23} \text{ atoms/mol}$$

$$\frac{N_V}{N} = \exp\left(\frac{-Q_V}{kT}\right) = 2.7 \times 10^{-4}$$

Annotations for the equation above:

- N_V (blue) points to the numerator of the fraction.
- N (black) points to the denominator of the fraction.
- Q_V (red) points to the red Q_V in the numerator of the exponent.
- kT (green) points to the green kT in the denominator of the exponent.
- 0.9 eV/atom (red) points to the red Q_V .
- 1273K (yellow) points to the yellow T in the denominator of the exponent.
- $8.62 \times 10^{-5} \text{ eV/atom-K}$ (green) points to the green k in the denominator of the exponent.

$$\text{For } 1 \text{ m}^3, N = \rho \times \frac{N_A}{A_{\text{Cu}}} \times 1 \text{ m}^3 = 8.0 \times 10^{28} \text{ sites}$$

- Answer:

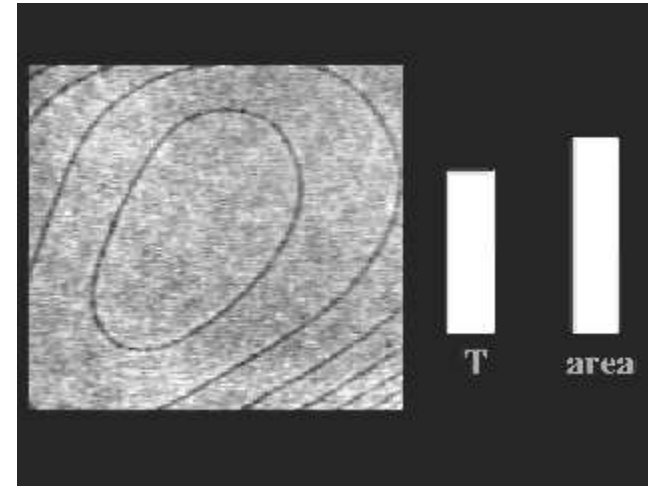
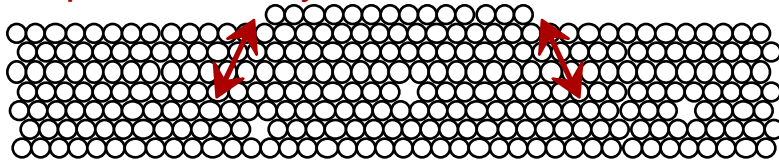
$$N_V = (2.7 \times 10^{-4})(8.0 \times 10^{28}) \text{ sites} = 2.2 \times 10^{25} \text{ vacancies}$$



Observing Equilibrium Vacancy Conc.

- Low energy electron microscope view of a (110) surface of NiAl.
- Increasing T causes surface island of atoms to grow.
- Why? The equil. vacancy conc. increases via atom motion from the crystal to the surface, where they join the island.

Island grows/shrinks to maintain equil. vacancy conc. in the bulk.

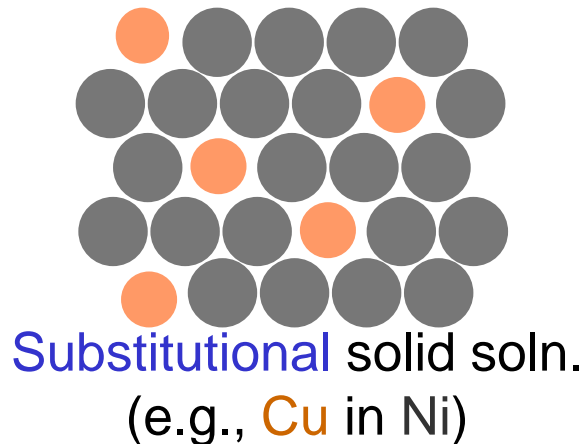


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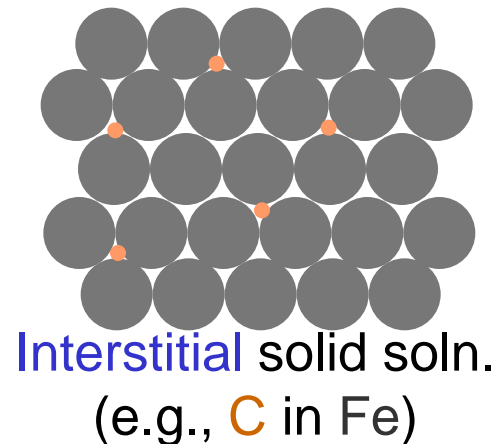
Point Defects in Alloys

Two outcomes if impurity (B) added to host (A):

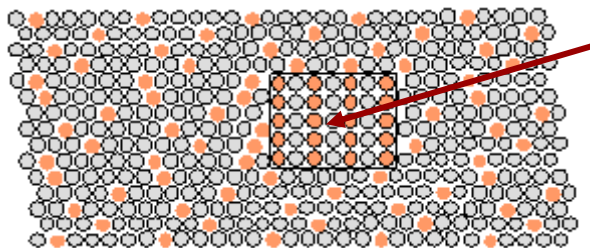
- **Solid solution** of B in A (i.e., random dist. of point defects)



OR



- Solid solution of B in A plus particles of a new phase (usually for a larger amount of B)



Second phase particle
--different **composition**
--often different structure.

Imperfections in Solids

Conditions for substitutional solid solution (S.S.)

- W. Hume – Rothery rule

- 1. Δr (atomic radius) < 15%
- 2. Proximity in periodic table
 - i.e., similar electronegativities
- 3. Same crystal structure for pure metals
- 4. Valency
 - All else being equal, a metal will have a greater tendency to dissolve a metal of higher valency than one of lower valency



Imperfections in Solids

Application of Hume–Rothery rules – Solid Solutions

1. Would you predict more Al or Ag to dissolve in Zn?

2. More Zn or Al in Cu?

<i>Element</i>	<i>Atomic Radius (nm)</i>	<i>Crystal Structure</i>	<i>Electro-negativity</i>	<i>Valence</i>
Cu	0.1278	FCC	1.9	+2
C	0.071			
H	0.046			
O	0.060			
Ag	0.1445	FCC	1.9	+1
Al	0.1431	FCC	1.5	+3
Co	0.1253	HCP	1.8	+2
Cr	0.1249	BCC	1.6	+3
Fe	0.1241	BCC	1.8	+2
Ni	0.1246	FCC	1.8	+2
Pd	0.1376	FCC	2.2	+2
Zn	0.1332	HCP	1.6	+2

Table on p. 106, Callister 7e.



Imperfections in Solids

- Specification of composition

- weight percent $C_1 = \frac{m_1}{m_1 + m_2} \times 100$

m_1 = mass of component 1

- atom percent $C'_1 = \frac{n_{m1}}{n_{m1} + n_{m2}} \times 100$

n_{m1} = number of moles of component 1



Line Defects

Dislocations:

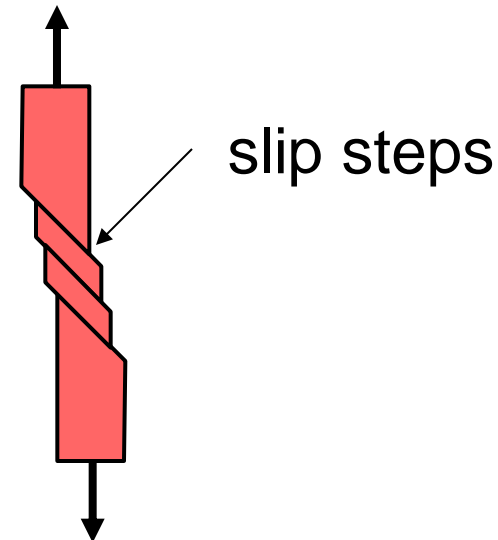
- are line defects,
- slip between crystal planes result when dislocations move,
- produce permanent (plastic) deformation.

Schematic of Zinc (HCP):

- before deformation



- after tensile elongation



Adapted from Fig. 7.8, *Callister 7e*.



Imperfections in Solids

Linear Defects (Dislocations)

- Are one-dimensional defects around which atoms are misaligned
- Edge dislocation:
 - extra half-plane of atoms inserted in a crystal structure
 - $\mathbf{b} \perp$ to dislocation line
- Screw dislocation:
 - spiral planar ramp resulting from shear deformation
 - $\mathbf{b} \parallel$ to dislocation line

Burger's vector, \mathbf{b} : measure of lattice distortion



Imperfections in Solids

Edge Dislocation

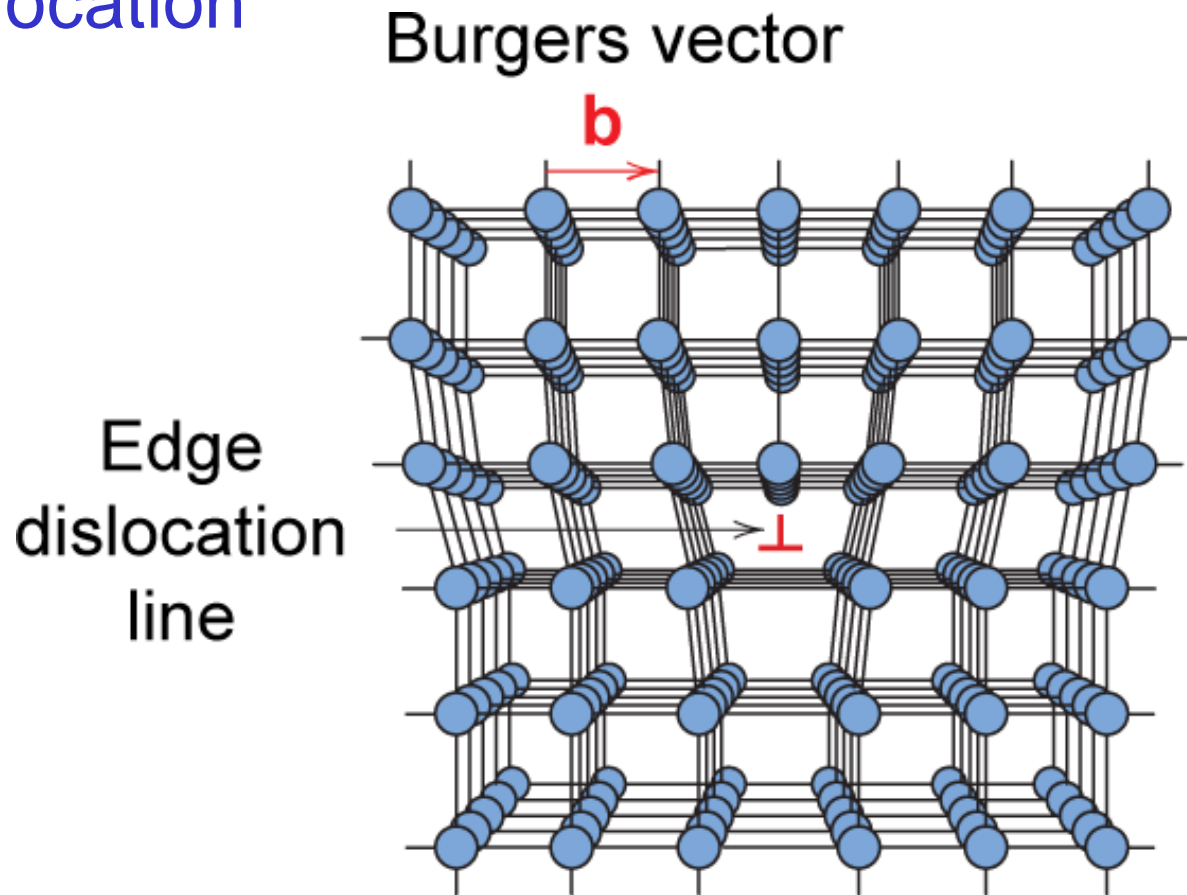
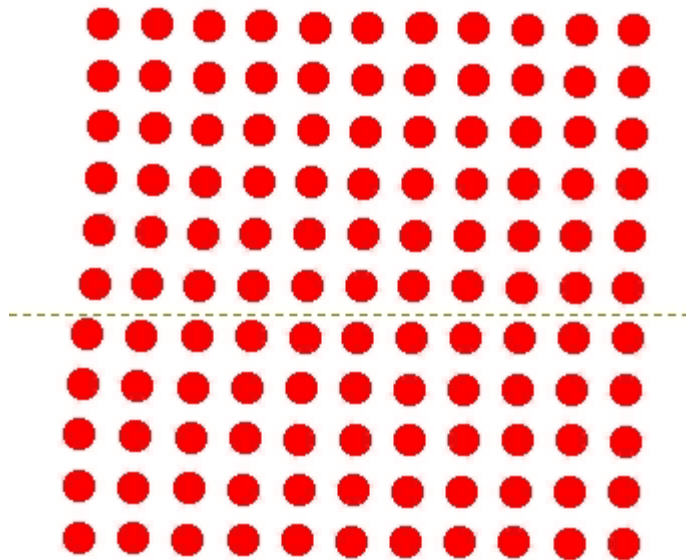


Fig. 4.3, Callister 7e.

Motion of Edge Dislocation

- Dislocation motion requires the successive bumping of a half plane of atoms (from left to right here).
- Bonds across the slipping planes are broken and remade in succession.



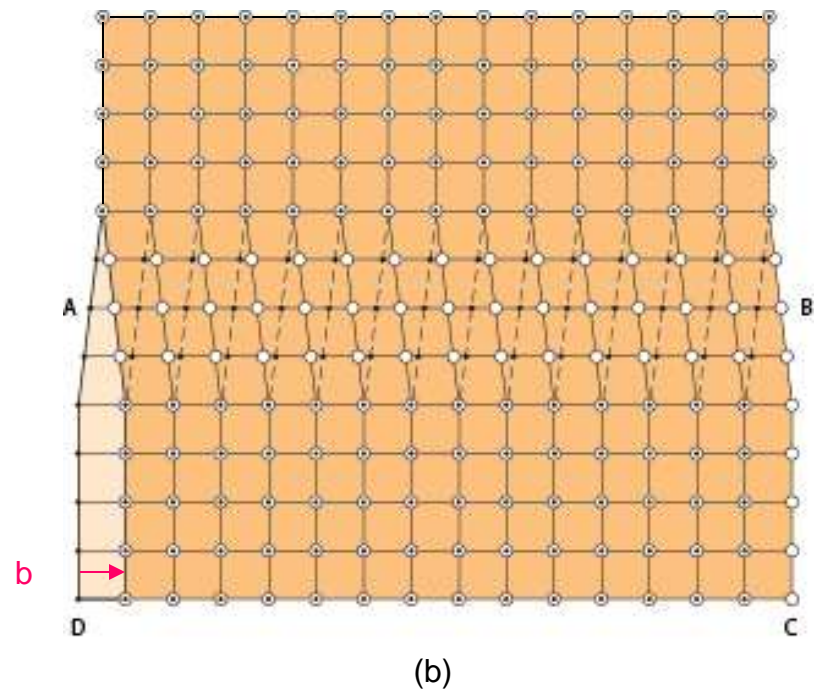
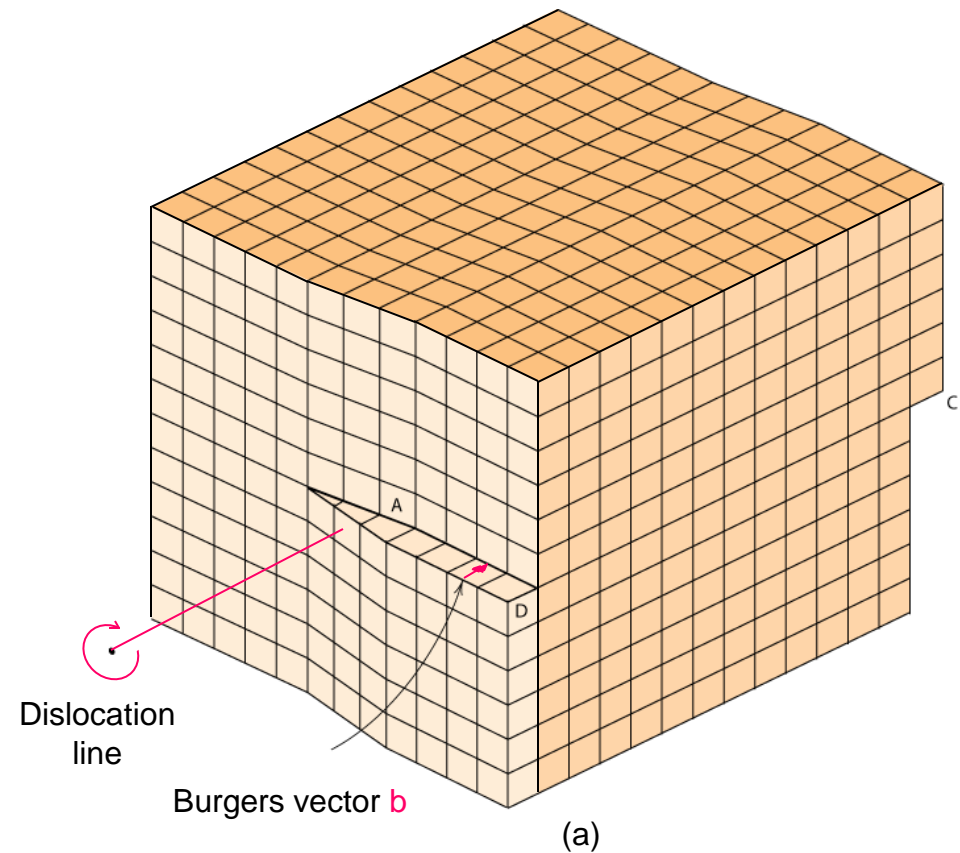
Atomic view of edge dislocation motion from left to right as a crystal is sheared.

(Courtesy P.M. Anderson)



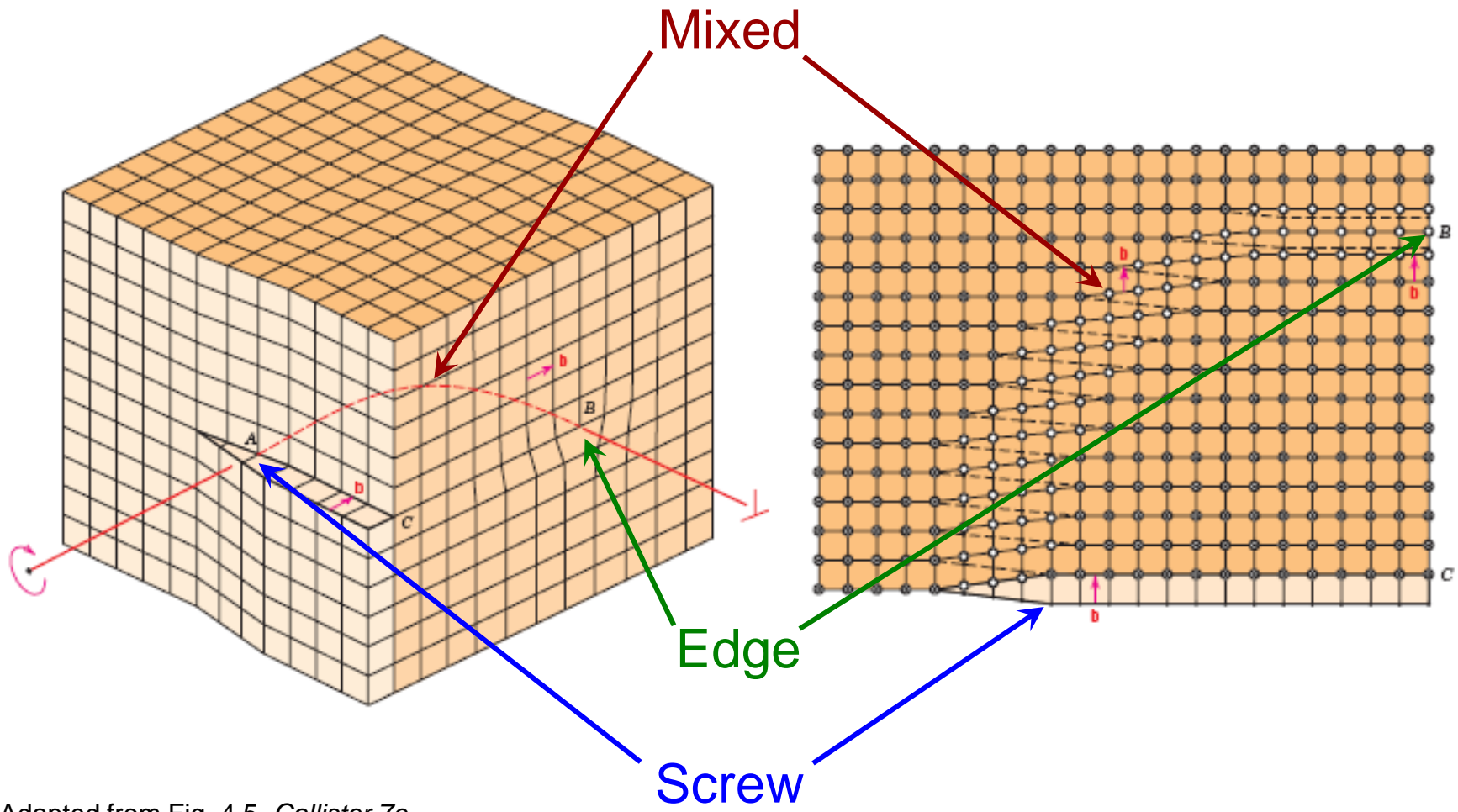
Imperfections in Solids

Screw Dislocation



Adapted from Fig. 4.4, *Callister 7e*.

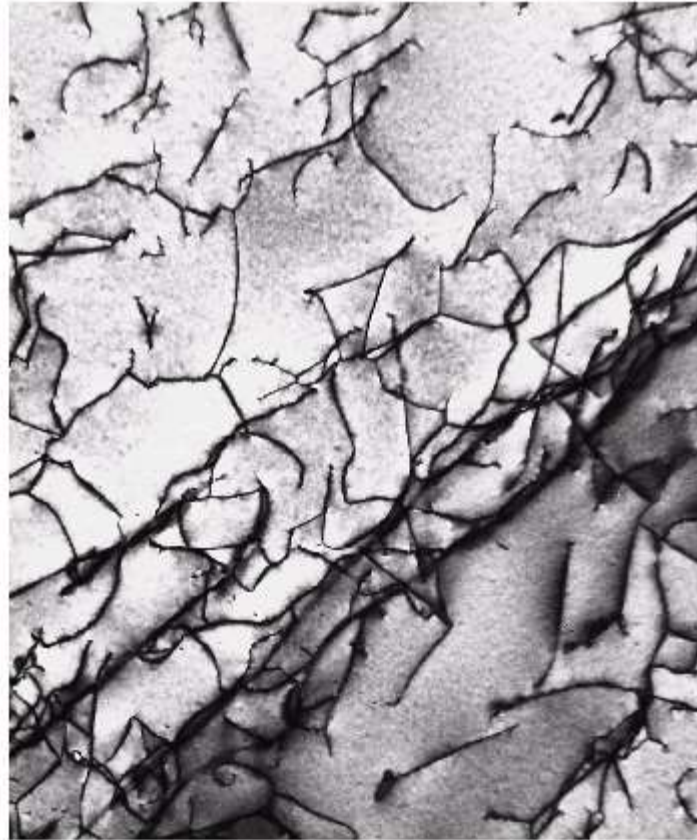
Edge, Screw, and Mixed Dislocations



Adapted from Fig. 4.5, Callister 7e.

Imperfections in Solids

Dislocations are visible in electron micrographs

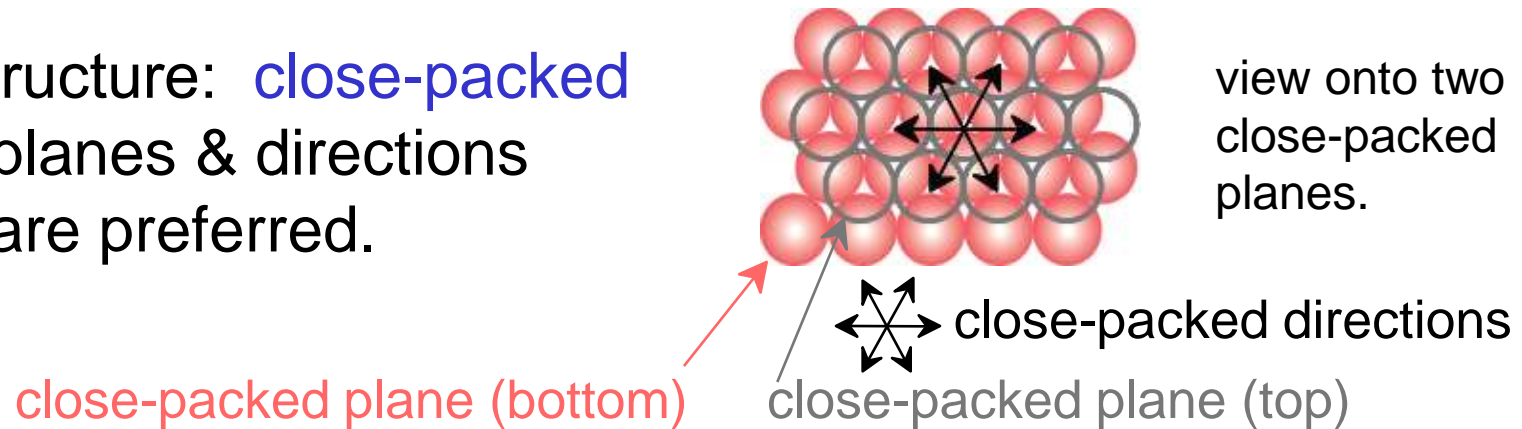


Adapted from Fig. 4.6, *Callister 7e*.



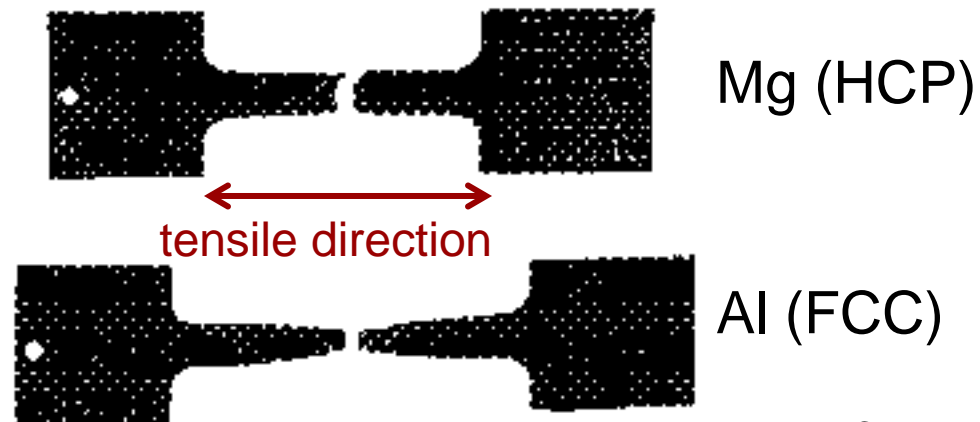
Dislocations & Crystal Structures

- Structure: **close-packed** planes & directions are preferred.



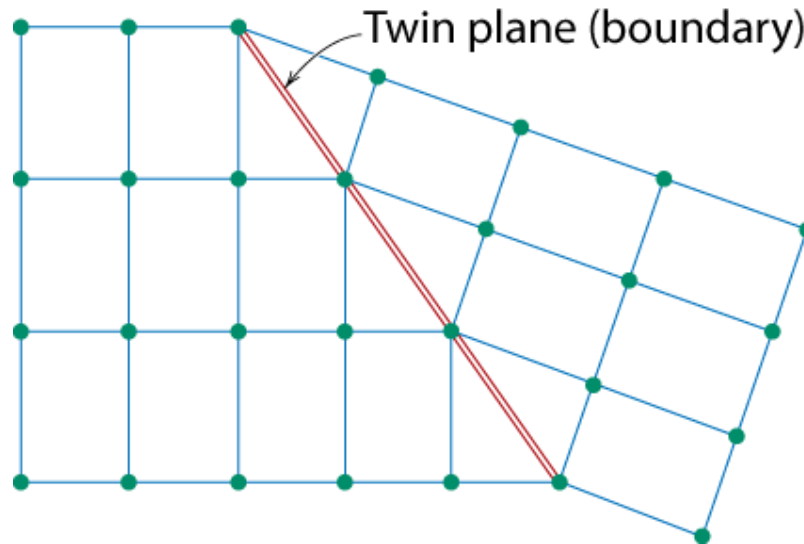
- Comparison among crystal structures:
FCC: many close-packed planes/directions;
HCP: only one plane, 3 directions;
BCC: none

- Specimens that were tensile tested.



Planar Defects in Solids

- One case is a **twin boundary (plane)**
 - Essentially a reflection of atom positions across the **twin plane**.



Adapted from Fig. 4.9, *Callister 7e*.

- **Stacking faults**
 - For FCC metals an error in ABCABC packing sequence
 - Ex: ABCABABC

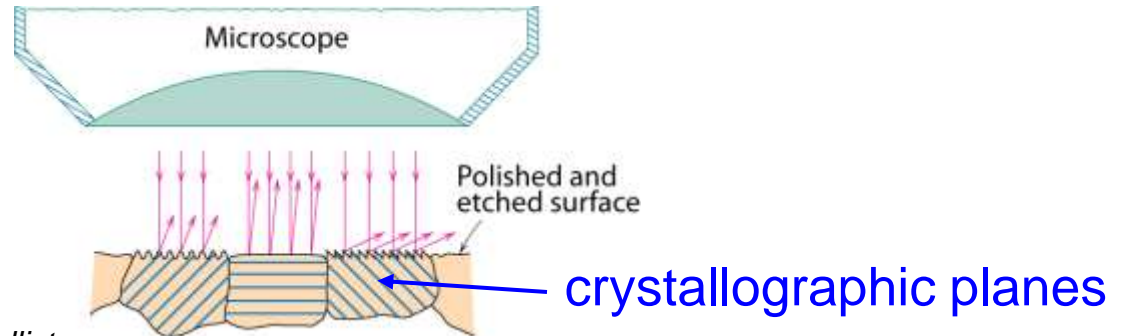
Microscopic Examination

- Crystallites (grains) and grain boundaries.
Vary considerably in size. Can be quite large
 - ex: Large single crystal of quartz or diamond or Si
 - ex: Aluminum light post or garbage can - see the individual grains
- Crystallites (grains) can be quite small (mm or less) – necessary to observe with a microscope.



Optical Microscopy

- Useful up to 2000X magnification.
- Polishing removes surface features (e.g., scratches)
- Etching changes reflectance, depending on crystal orientation.



Adapted from Fig. 4.13(b) and (c), *Callister 7e*. (Fig. 4.13(c) is courtesy of J.E. Burke, General Electric Co.



Micrograph of brass (a Cu-Zn alloy)

← 0.75mm →



Optical Microscopy

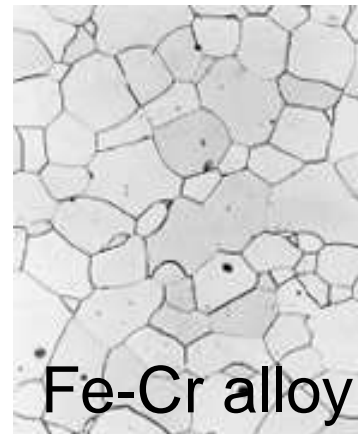
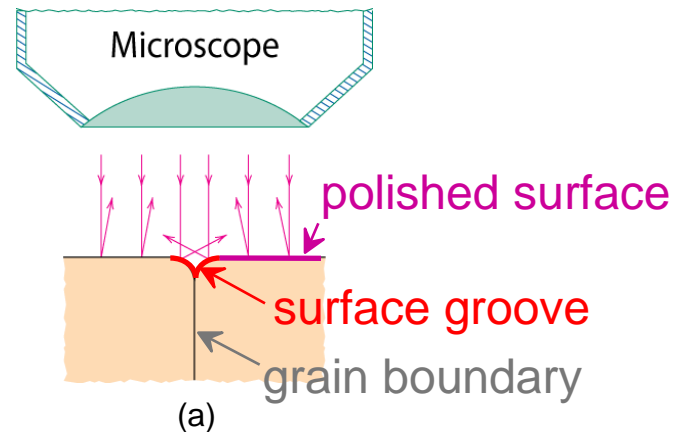
Grain boundaries...

- are imperfections,
- are more susceptible to etching,
- may be revealed as dark lines,
- change in crystal orientation across boundary.

ASTM grain size number

$$N = 2^{n-1}$$

number of grains/in²
at 100x
magnification



(b)

Adapted from Fig. 4.14(a) and (b), *Callister 7e*.
(Fig. 4.14(b) is courtesy of L.C. Smith and C. Brady, the National Bureau of Standards, Washington, DC [now the National Institute of Standards and Technology, Gaithersburg, MD].)



Optical Microscopy

- Polarized light
 - metallographic scopes often use polarized light to increase contrast
 - Also used for transparent samples such as polymers



Microscopy

Optical resolution ca. 10^{-7} m = 0.1 μ m = 100 nm

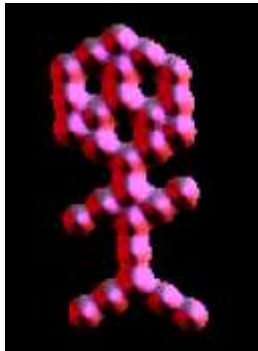
For higher resolution need higher frequency

- X-Rays? Difficult to focus.
- Electrons
 - wavelengths ca. 3 pm (0.003 nm)
 - (Magnification - 1,000,000X)
 - Atomic resolution possible
 - Electron beam focused by magnetic lenses.

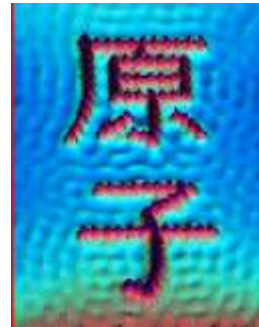


Scanning Tunneling Microscopy (STM)

- Atoms can be arranged and imaged!



Carbon monoxide molecules arranged on a platinum (111) surface.



Iron atoms arranged on a copper (111) surface. These Kanji characters represent the word “atom”.

Photos produced from the work of C.P. Lutz, Zeppenfeld, and D.M. Eigler. Reprinted with permission from International Business Machines Corporation, copyright 1995.



Summary

- Point, Line, and Area defects exist in solids.
- The number and type of defects can be varied and controlled (e.g., T controls vacancy conc.)
- Defects affect material properties (e.g., grain boundaries control crystal slip).
- Defects may be desirable or undesirable (e.g., dislocations may be good or bad, depending on whether plastic deformation is desirable or not.)



ANNOUNCEMENTS

Reading:

Core Problems:

Self-help Problems:

